3-2011

Effect of hysteresis on measurements of thin-film cell performance

David S. Albin
National Renewable Energy Laboratory

Joseph A. del Cueto
National Renewable Energy Laboratory

Follow this and additional works at: https://digitalscholarship.unlv.edu/renew_pubs
Part of the Oil, Gas, and Energy Commons, and the Power and Energy Commons

Repository Citation
Available at: https://digitalscholarship.unlv.edu/renew_pubs/50

This Conference Proceeding is brought to you for free and open access by the Energy at Digital Scholarship@UNLV. It has been accepted for inclusion in Publications (E) by an authorized administrator of Digital Scholarship@UNLV. For more information, please contact digitalscholarship@unlv.edu.
Effect of Hysteresis on Measurements of Thin-Film Cell Performance

David Albin and Joseph del Cueto

*Presented at the Reliability of Photovoltaic Cells, Modules, Components, and Systems III: SPIE Conference*  
*San Diego, California*  
*August 3-5, 2010*
Notice

The submitted manuscript has been offered by an employee of the Alliance for Sustainable Energy, LLC (Alliance), a contractor of the US Government under Contract No. DE-AC36-08GO28308. Accordingly, the US Government and Alliance retain a nonexclusive royalty-free license to publish or reproduce the published form of this contribution, or allow others to do so, for US Government purposes.

This report was prepared as an account of work sponsored by an agency of the United States government. Neither the United States government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States government or any agency thereof.

Available electronically at http://www.osti.gov/bridge

Available for a processing fee to U.S. Department of Energy and its contractors, in paper, from:

U.S. Department of Energy
Office of Scientific and Technical Information

P.O. Box 62
Oak Ridge, TN 37831-0062
phone: 865.576.8401
fax: 865.576.5728
email: reports@adonis.osti.gov

Available for sale to the public, in paper, from:

U.S. Department of Commerce
National Technical Information Service
5285 Port Royal Road
Springfield, VA 22161
phone: 800.553.6847
fax: 703.605.6900
email: orders@ntis.fedworld.gov
online ordering: http://www.ntis.gov/help/ordermethods.aspx

Cover Photos: (left to right) PIX 16416, PIX 17423, PIX 16560, PIX 17613, PIX 17436, PIX 17721

Printed on paper containing at least 50% wastepaper, including 10% post consumer waste.
Effect of Hysteresis on Measurements of Thin-Film Cell Performance

David Albin* and Joseph del Cueto
National Renewable Energy Laboratory, 1617 Cole Blvd., Denver, CO, USA 80401

ABSTRACT

Transient or hysteresis effects in polycrystalline thin film CdS/CdTe cells are a function of pre-measurement voltage bias and whether Cu is introduced as an intentional dopant during back contact fabrication. When Cu is added, the current-density (J) vs. voltage (V) measurements performed in a reverse-to-forward voltage direction will yield higher open-circuit voltage ($V_{oc}$), up to 10 mV, and smaller short-circuit current density ($J_{sc}$), by up to 2 mA/cm$^2$, relative to scanning voltage in a forward-to-reverse direction. The variation at the maximum power point, $P_{max}$, is however small. The resulting variation in FF can be as large as 3%. When Cu is not added, hysteresis in both $V_{oc}$ and $J_{sc}$ is negligible however $P_{max}$ hysteresis is considerably greater. This behavior corroborates observed changes in depletion width, $W_d$, derived from capacitance (C) vs voltage (V) scans. Measured values of $W_d$ are always smaller in reverse-to-forward voltage scans, and conversely, larger in the forward-to-reverse voltage direction. Transient ion drift (TID) measurements performed on Cu-containing cells do not show ionic behavior suggesting that capacitance transients are more likely due to electronic capture-emission processes. J-V curve simulation using Pspice shows that increased transient capacitance during light-soak stress at 100 °C correlates with increased space-charge recombination. Voltage-dependent collection however was not observed to increase with stress in these cells.

Keywords: CdTe solar cell, transient capacitance, DLTS, transient ion drift, reliability, degradation, voltage-dependent collection

1. INTRODUCTION

Transient and metastable effects in polycrystalline thin film modules significantly affect module performance measurements. Storage conditions concerning light-exposure, bias, and temperature can change not only power, but also measurements of open-circuit voltage ($V_{oc}$), short-circuit current-density ($J_{sc}$), and fill-factor (FF) [1]. The magnitude of these effects is observed to vary as modules undergo accelerated lifetime testing [2]. Uncertainty regarding pre-measurement conditions during module shipment and storage and the resulting effect on performance justifies the need for stabilization procedures prior to module performance measurements during qualification and field-test studies.

Recent research has concentrated on studying the nature of transients in capacitance-voltage (C-V) measurements applied to small-area CdTe devices [3] as well as production-sized CdTe and Cu(In,Ga)Se$_2$ modules [2]. These studies involve measuring the capacitance as a function of sequential forward-to-reverse (termed a “rev” direction) and reverse-to-forward (termed a “fwd” direction) voltage scans with a 5 m hold at reverse-bias. In [3], CdS/CdTe devices were fabricated on two, uniquely different transparent conducting oxides (TCOs) in which the durability of the device under 1-sun, $V_{oc}$ bias, 100 °C stress testing was notably affected by the TCO type. The rate of degradation observed in CdS/CdTe devices fabricated on insulating zinc-stannate (ZTO)/cadmium stannate (CTO) TCO/Corning 7059 borosilicate glass substrates was in general, greater relative to identically fabricated CdS/CdTe devices fabricated on more traditional, undoped, insulating tin-oxide (iSnO$_2$)/F-doped conductive tin-oxide (cSnO$_2$) TCO/borosilicate glass substrates. Depletion width ($W_d = \epsilon \cdot A)/C$ where $\epsilon$ is the dielectric permittivity, A is the cell area) hysteresis (defined as the difference in $W_d$ measured at V=0 between fwd and rev voltage scans) was observed to correlate strongly with performance during stress testing. The variation in $W_d$ hysteresis with stress time as well as the correlation of this capacitance-derived measurement with both $V_{oc}$ and FF measured during stress is shown in Figure 1. The increased hysteresis seen in Fig. 1(a) paralleled decreased performance during stress. The reasonably good correlation of hysteresis with both $V_{oc}$ and FF (shown in Fig. 1(b) and 1(c) respectively) is also apparent.
The linear correlation coefficient, $R^2$, of $V_{oc}$ with hysteresis for CTO/ZTO cells #1 and #2, and SnO$_2$ cells #1 and #2, were 0.98, 0.46, 0.75, and 0.82, respectively. Similar correlations for FF were 0.99, 0.58, 0.63, and 0.87 respectively. When analyzed without considering hysteresis, i.e., based on either $W_{d,fwd}$ or $W_{d,rev}$ separately, correlations were not nearly as high and tended to show non-monotonic behavior.

The origin of this capacitance hysteresis was not substantiated. From an electronic perspective, hysteresis could be explained by considering majority carrier trap emission within the CdTe space-charge region. Depletion widths measured during a rev-voltage scan will initially be higher since the negative space-charge (i.e., $N_a^-$) region of the CdTe is screened by holes occupying deep states in the band gap. At a fixed reverse bias (1.5 V in [3]), some deep states fall below the Fermi level and the resulting, time-dependent hole emission (or electron capture) will reduce this screening causing $W_d$ to decrease. The subsequent fwd-direction voltage scan (-1.5 V to +0.2 V) captures this reduced value of $W_d$ such that hysteresis, $W_{d,rev} - W_{d,fwd}$ is positive. This filling and emptying of deep states gives rise to capacitance transients which are fundamental to deep level transient spectroscopy (DLTS) measurements. Based upon this model, hysteresis is proportional to the density of deep states within the CdTe space-charge undergoing capture-emission processes, and an increase in hysteresis would represent an increase in electronic traps with stress.

Another mechanism for hysteresis, based upon transient ion drift (TID) was also presented. This seemed plausible given the previous reports of TID behavior in both polycrystalline CdTe [4] and CuInSe$_2$ [5] devices which themselves were based upon earlier works by Heiser and Mesli [6] and subsequently, Lyubomirsky, et al. [7]. In the TID model for hysteresis, the higher $W_d$ measured during the rev-voltage scan is due to the screening of the space-charge region, not by holes, but rather by positively charged Cu-interstitial donors, Cu$_i^+$. Again, at a fixed reverse bias of 1.5 V, these donors are attracted toward the back contact, away from the space-charge, with the same net effect, i.e., a decrease in $W_d$ again.
due to reduced screening. However, according to Heiser and Mesli, the transients observed in TID are distinguishable from those in TID. Referring to the capacitance change occurring at reverse and positive bias as accumulation (Cu⁺ accumulating towards the back contact) and flattening (Cu⁺ re-distribution towards the space-charge) processes respectively, they note that relaxation times at reverse and forward bias levels, τₐ and τᵣ, should follow the relationship τᵣ/τₐ > 1 and be reverse bias dependent, that is, the ratio should increase with increasing reverse-bias. This was argued as not the case for DLTS processes where the capture rate is much higher than the emission rate, and generally independent of reverse bias. The C-V technique described in [3] unfortunately did not record decay rates at either +0.2 or -1.5 V. It was stated however, that there was no obvious mechanism by which an increase in mobile ions (as explanation for the increased hysteresis shown in Fig. 1) would degrade performance, but rather, the formation of deep states and possibly increased recombination that would most likely decrease performance during stress testing.

This paper reports on a modification of the C-V technique used in [3] such that capacitance response is measured during the initial reverse-bias pulse (τₐ) and a subsequent, relaxation bias at V=0 (τᵣ). By repeating this measurement at progressively increasing reverse-bias levels, the voltage-dependence of the term τᵣ/τₐ can be determined. The effect of the strong variation in W₄ with voltage-scan direction shown in Fig. 1(a) on the J-V characteristics of cells with and without Cu as an intentional dopant is also shown. Finally, further insight regarding the mechanism of degradation reported in [3] is ascertained using PSpice modeling of device J-V characteristics measured during stress.

2. EXPERIMENTAL PROCEDURE

Cell Fabrication

The structure and fabrication history of CdS/CdTe cells used in this study is discussed in reference [3]. The basic structure is that of a Corning 7059 borosilicate glass superstrate design in which light passes through a glass/TCO/buffer construction (either 7059/cSnO₂/iSnO₂ or 7059/CTO/ZTO), where it is then absorbed in the n-(CdS)/p-(CdTe) heterojunction. A schematic of the different layers comprising the device is shown in Fig. 2.

![Figure 2. CdS/CdTe device superstrate structure.](image)

The CTO/ZTO and SnO₂-based TCOs were deposited by sputtering and low-pressure, chemical vapor deposition. The cSnO₂ and iSnO₂ layers were deposited onto bare glass substrates heated to 550 °C using tetramethyltin, bromotrifluoromethane (CBrF₃), and oxygen. The CBrF₃ provides the fluorine dopant for the cSnO₂ layer alone. The cSnO₂ and iSnO₂ layers were 500 and 100 nm thick respectively. The resulting sheet resistance of this bi-layer stack measures ~ 9 Ω/sq. The CTO and ZTO layers were sputtered onto unheated substrates to a thickness of 320 and 150 nm respectively. After each sputter deposition, the amorphous oxide layers were annealed in 30 Torr He for 15 min at 650 °C with the oxide film in contact with a CdS film/Corning 7059 glass plate (i.e., a “close-proximity” anneal). During the subsequent anneal, the optical band gap of the CTO layer increases from about 3.0 eV to 3.5 eV with a several order
magnitude drop in resistivity to $\sim 1.8 \times 10^{-4}$ Ω·cm. The CTO layer exhibits some crystallization during this post-deposition anneal. The iSnO$_2$ and ZTO buffer layers (after anneal) are insulating with a resistivity of around 1-10 Ω·cm and contribute little additional absorption loss to the oxide layer stack. The CdS layer is grown by chemical bath deposition at ~90 °C to a thickness of approximately 80 nm. The CdTe layer is deposited onto the CdS/buffer/TCO/glass substrate in a close-spaced sublimation (CSS) arrangement where the CdTe source and substrate temperatures are 660 and 625 °C respectively. An ambient of 15 torr He and 1 torr O$_2$ is used during CSS. After the CdTe deposition, the CdTe/CdS/buffer/TCO/glass structure was then exposed to vapor CdCl$_2$ treatments in an oxygen/helium ambient (100 Torr O$_2$ + 400 Torr He). Prior to the back contact application, CdTe surfaces were etched in a nitric-phosphoric acid solution in order to form a beneficial, Te-rich layer and to remove surface oxides [8]. Back contacts concluded with the application of a relatively thick layer (~50–100 µm) of graphite paste containing Cu$_x$Te and HgTe dopants followed by a similarly thick, final conducting layer of Ag paste. Cells without intentional Cu dopant use only the graphite paste. A narrow (~1 mm) margin of CdTe surrounding the perimeter of the dopant/Ag paste contact was used to reduce edge shunting.

**J-V and C-V Measurements**

Performance data using standard current versus voltage (J-V) scans were made on cells after fabrication ($t = 0$) and during stress testing ($t > 0$) with a current-calibrated Oriel solar simulator. In an attempt to average out J-V transients due to pre-bias and voltage scan directions, the voltage applied across the probes is routinely swept continuously from a fwd-bias to rev-bias in a saw-tooth fashion while Kelvin probes are lowered onto the front and back contacts of the device. With probes attached, and after the next subsequent voltage sweep, current as a function of voltage is collected first in the dark from fwd to rev-bias. The simulator shutter is then opened, exposing the cell to a pre-calibrated 1-sun irradiance spectrum with the photocurrent measured during the subsequent fwd to rev-scan. In this fashion, transients are managed and repeatable measurements of performance are obtained for monitoring relative degradation during stress testing. A small indium pad is placed between the back contact and Kelvin probe to minimize potential damage to the cell by the sharp probe tips. To ascertain what effect the hysteresis in depletion width shown in Fig. 1(a) has on cell performance, this voltage sweeping technique was not used. Rather, cells were first biased at -1.5 V for 5 min in the dark, at which point the current was measured as the voltage was swept from -1.5 V to +0.2 V. At +0.2 V, the cell was then placed in the dark for an additional 5 m, and the light photocurrent was measured as the voltage was scanned from +0.2 V to -1.5 V.

Capacitance-voltage (C-V) measurements (dark at room temperature) were performed using an Agilent 4294A Precision Impedance Analyzer operated at 100 kHz with a 50 mV oscillation voltage. The capacitance is relatively insensitive to frequency at 100 kHz. Unlike in [3], a Labview program was used to automate data collection to insure repeatability. The same bi-directional voltage scan used in [3] was replicated except that three successive loops in voltage between $V=0$ and increasing reverse biases of -1.0, -2.0, and -3.0 V were performed with capacitance transient data collected at the termination of each voltage scan. After some experimentation, a pause time of 20 m was used to collect transient data at each terminating voltage. This is twice as long as the collection time used in reference [4] and was determined to be adequate for obtaining linear fits of $\ln [\Delta C]$ functions with time for calculating relaxation time constants. The relaxation time constants for $\tau_a$ and $\tau_f$ were determined respectively, using the following relationships:

$$\frac{C(\infty) - C(t)}{C(\infty) - C(0)} = \exp\left(-\frac{t}{\tau_a}\right)$$

$$\frac{C(t) - C(\infty)}{C(0) - C(\infty)} = \exp\left(-\frac{t}{\tau_f}\right)$$

where $C(t)$, $C(0)$, and $C(\infty)$ equal the capacitance as a function of time, as well as the capacitance at the beginning and end of each 20 m transient measurement period. The resulting capacitance change with time for a representative Cu-containing cell is shown in Fig. 3.
Figure 3. Capacitance versus time for a three cycle loop with progressively increasing rev-bias voltage.

The increasing capacitance during the accumulation period, where the cell is subjected to a reverse bias pulse, corresponds to a value for \( W_d \) which decreases with time (\( C \) increases). \( C(0) \) and \( C(\infty) \) in equation (1) correspond to the capacitance measured at \( t = 0 \) and 1200 s (20 m). The voltage is then swept to \( V = 0 \) capturing the hysteresis discussed in [3]. The voltage is maintained at this level for an additional 20 m where \( W_d \) increases (\( C \) decreases). \( C(0) \) and \( C(\infty) \) in equation (2) corresponds to the capacitance measured at \( t=1200^\circ \) and 2400 s.

Accelerated lifetime test (ALT) conditions have been described before [3]. In brief, cells were placed, glass-side up, under an Atlas CPS+ solar light source (~AM 1.5; 1-sun) in machined Al blocks designed to keep cells at \( V_{oc} \) bias. Cell temperature was set to 100 °C. At times equal to 1, 4.4, 10, 28, 73, and 115 hrs, cells were removed and allowed to stabilize in the dark for 12-24 hrs. After measurements of \( J-V \) and \( C-V \), cells were again placed under stress. Sometime after the last measurement at \( t=115 \) hrs, the temperature controllers failed (over a weekend) resulting in excessive heating and destruction of cells. The actual test temperature of 100 °C is suspect (temperatures may have been higher) particularly near the conclusion of the test. Regardless, the design of the Al blocks at least assures us that cells were tested at the same temperature.

**J-V Curve Modeling**

PSpice simulations provide a near-quantitative method for determining losses in devices as a function of stress. In Fig. 4, a parallel combination of forward-biased diodes is used to independently simulate recombination currents in the quasi-neutral (\( J_{QNR} \)) and space-charge (\( J_{SCR} \)) regions. Back contact behavior is represented by a parallel combination of a reverse-biased diode (represented by the leakage current \( J_b \)) and a shunt conductance (\( 1/R_b \)) that was previously used to model the \( J-V \) characteristics of contacts commonly observed in 1st quadrant \( J-V \) behavior [9]. The theoretical basis for the two-diode model as a general expression for the current produced in a solar cell can be found in reference [10].

![Two-diode solar cell discrete element circuit model](image)

Figure 4. Two-diode solar cell discrete element circuit model.
Losses associated with any of the circuit elements shown in Fig. 4 can be determined by comparing J-V curves before and after some change in the cell. For example, Fig. 5 demonstrates a nearly exact fit for a 14.4% cell in which the following parameters were determined: $J_p = 0.024409 \text{ A/cm}^2$, $J_{01} = 0.5 \times 10^{-16} \text{ A/cm}^2$, $J_{02} = 2 \times 10^{-9} \text{ A/cm}^2$, $R_s = 1.9 \text{ ohm}\cdot\text{cm}^2$, $R_{sh} = 200k \text{ ohm}\cdot\text{cm}^2$, $J_b = 6 \times 10^{-2} \text{ A/cm}^2$, and $R_b = 1.5 \text{ ohm}\cdot\text{cm}^2$. The “loss” associated with this cell, relative to a theoretical, maximum performance cell can be determined by generating the latter J-V curve as shown in Fig. 5 in which $J_{01}$ and $J_{02}$ are both set to ideal values of $1 \times 10^{-20}$, $R_s \rightarrow 0$, $R_{sh} \rightarrow \infty$, and $J_b$ and $R_b$ are removed from the model. The corresponding ideal cell has J-V parameters of $V_{oc} \sim 1.095 \text{ V}$, $J_{sc} \sim 24.4 \text{ mA/cm}^2$, FF $\sim 89.0$, and efficiency of $\sim 23.8\%$ suggesting the performance entitlement of this technology if processing induced “losses”, in particular those resulting in high recombination currents can be eliminated. This approach has recently been used to explain general degradation mechanisms in CdS/CdTe devices as a function of stress [11].

![Figure 5](image_url)

**Figure 5.** PSpice simulation of a high-efficiency CdS/CdTe device and the potential improvement in J-V behavior if all losses are removed (ideal cell).

A major weakness of PSpice, however, is its inability to adequately determine voltage-dependent collection losses since $J_p$ in the PSpice model is constant, and not a function of voltage. Thus, superposition is guaranteed. The high degree of superposition between the light and dark J-V curves for the ideal cell shown in Fig. 5 was obtained simply by setting $J_p=0$ in the dark J-V simulation. In general, polycrystalline thin film solar cells do not demonstrate this degree of superposition [12]. However, as discussed in reference [11], the differences between actual J-V data and the PSpice simulations can be used to determine this voltage-dependent photocurrent loss. The value of $J_p$ is set to the current measured at the greatest value of reverse-bias. $R_s$ is then determined by $dV/dJ$ in the first quadrant. As long as roll-over is not extreme ($J_b$ approaching zero), this value can be determined. $R_{sh}$ is subsequently determined by using the value of $dV/dJ$ at $V=0$ from the dark J-V curve since there is nothing to suggest a photo-active shunt conductance. The remaining parameters $J_{QNR}$ and $J_{SCR}$ are then adjusted to fit $V_{oc}$. When performed in this manner, the actual J-V photocurrent will generally be lower than the simulated value since the latter again assumes a constant value of $J_p$. The difference between these two curves represents the voltage-dependent loss.

### 3. RESULTS AND DISCUSSION

**Pre-bias Effects on Cell Performance**

Fig. 6 shows how pre-biasing the cell at either 1.5 V reverse bias and 0.2 V forward bias can affect the J-V characteristics of cells with and without intentional Cu added as a dopant to the graphite paste contact described previously. The behavior shown in Fig. 6 is representative of the 16 cells (8 each with and without Cu) fabricated for this measurement.
Figure 6. Representative variation in cell J-V curve behavior based upon pre-measurement bias and sweep direction for CdS/CdTe cells without (a) and with (b) intentional Cu added as a dopant.

$V_{oc}$ and $J_{sc}$ were strongly impacted by pre-measurement bias and sweep direction only for devices that contained Cu. At the same time, the behavior at $P_{max}$ (where J-V curves cross) was significantly less. Scanning the cell in a fwd direction from an initial reverse bias — where C-V data would suggest a smaller $W_d$ — caused a strong reduction in $J_{sc}$ (1-2 mA/cm²) and increase in $V_{oc}$ (5-10 mV). The net effect on FF is an increase of 1-3% points. This is similar to the increase in FF observed in testing CdTe modules when tracing current-voltage from $V=0$ ($I_{sc}$) to $V_{oc}$ (see for example Fig. 5 in reference [1]). This overall behavior is consistent in devices where voltage-dependent collection limits performance. $V_{oc}$ and $J_{sc}$ transients for cells not containing Cu were negligible, though a slight J-V curve hysteresis near $P_{max}$ was observed. FF was slightly less in cells not containing Cu when scanning voltage in the fwd direction. It should be noted that the distinction between cells “containing” and “not containing” Cu refers only to whether Cu was intentionally added during back contact processing. Cu may exist as a trace impurity in cells not intentionally doped with Cu, and thus the reason for the very slight hysteresis observed here in J-V measurements and in the C-V data presented earlier in [3].

TID measurements

Using the C-V technique described earlier to collect capacitance transient response at V=0 and progressively increasing values of reverse bias, equations (1) and (2) were employed to determine relaxation time constants in cells containing Cu. Capacitance transients in cells without Cu are too small to differentiate from noise in such measurements. Shown in Fig. 7 are the ln ($\Delta C$) functions plotted as a function of time and the corresponding relaxation time constants for accumulation ($\tau_a$) occurring during the reverse-bias pulse (7(a)) and flattening ($\tau_f$) at $V = 0$ (7(b)) for each of the three reverse-bias voltage levels studied. The latter values were determined from the inverse slope of the curves between $t = 200$ and 800 s. These measurements were performed for several cells containing Cu. The example shown here is representative of the overall behavior and will be used for this discussion. The ratio $\tau_f / \tau_a$ with increasing reverse bias is shown to equal 0.85, 0.79, and 0.72 in this case. This behavior does not meet either of the two primary criteria for identifying ionic drift as discussed previously. Thus, at this point, it does not appear that the hysteresis observed in these cells has an ionic basis though this work is somewhat preliminary. For example, frequency domain measurements might be used to selectively freeze out ionic processes.
Figure 7. Transient capacitance response during the accumulation (a) and flattening (b) processes occurring at reverse-bias and $V=0$ conditions respectively.

**J-V Curve Modeling**

J-V curves corresponding to cells CTO/ZTO #1 and SnO$_2$ #1 from Fig. 1 at $t = 0$ and $t = 115$ hrs were fit using the PSpice technique described previously. Actual and simulated J-V data for the CTO/ZTO and SnO$_2$ cells are shown in Fig. 8(a) and 8(b) respectively. The current density axis has been adjusted to magnify the difference observed between simulated and actual photocurrents. The difference between actual and fit graphs represent the additional current loss not determined by the discrete element model of Fig. 4. In each case, $R_{sh}$ was determined by the behavior of the dark J-V curve at $V = 0$ (not shown).

![CTO/ZTO J-V Curve](image1)

![SnO$_2$ J-V Curve](image2)

Figure 8. Comparison of actual J-V data and PSpice simulated data for unstressed ($t=0$) and stressed ($t=115$ hrs) CTO/ZTO-based (a) and SnO$_2$-based (b) cells.

The circuit model fit parameters for the simulated J-V curves are summarized in Table 1. Times are given in hrs, current density is expressed in A/cm$^2$, and resistive terms given in $\Omega \cdot$cm$^2$. The primary reason for the greater rate of degradation in the CTO/ZTO cells appears to be due to both increased shunting (lower $R_{sh}$), and increased space-charge recombination. At the same time, voltage-dependent collection does not appear to increase significantly with stress for CdTe devices using either substrate type. Increased recombination is likely due to the increased deep state density during stress testing. The latter manifests as an increase in the capacitance hysteresis observed during stress. It is interesting to note that the increased concentration of deep states does not appear to significantly increase voltage-dependent collection.
### Table 1.

<table>
<thead>
<tr>
<th>Substrate</th>
<th>Stress Time</th>
<th>Jp</th>
<th>J(o1)</th>
<th>J(o2)</th>
<th>Rs</th>
<th>Rsh</th>
<th>Rb</th>
<th>Jb</th>
</tr>
</thead>
<tbody>
<tr>
<td>CTO/ZTO</td>
<td>0</td>
<td>0.024502</td>
<td>3.00E-16</td>
<td>8.00E-10</td>
<td>4.3</td>
<td>200K</td>
<td>0.5</td>
<td>1.00E-01</td>
</tr>
<tr>
<td>CTO/ZTO</td>
<td>115</td>
<td>0.025289</td>
<td>3.00E-16</td>
<td>5.70E-09</td>
<td>4.6</td>
<td>595</td>
<td>6.7</td>
<td>2.70E-02</td>
</tr>
<tr>
<td>iSnO2/SnO2</td>
<td>0</td>
<td>0.02273</td>
<td>2.50E-16</td>
<td>6.00E-10</td>
<td>2.4</td>
<td>200K</td>
<td>1.3</td>
<td>8.50E-02</td>
</tr>
<tr>
<td>iSnO2/SnO2</td>
<td>115</td>
<td>0.023104</td>
<td>2.50E-16</td>
<td>2.90E-09</td>
<td>2.6</td>
<td>200K</td>
<td>4.6</td>
<td>4.80E-02</td>
</tr>
</tbody>
</table>

The transient behavior observed in C-V measurements obtained on CdTe devices deposited on either SnO2 or CTO/ZTO based substrates is likely due to the presence of deep states in the space-charge region. The increase in C-V hysteresis with stress testing of these cells is likely due to an increase in deep state density. This increases space-charge recombination. Higher levels of hysteresis are correlated with higher levels of degradation in cells. The latter is correlated with higher levels of recombination. Increased recombination does not appear to increase voltage-dependent collection. TID measurements performed as part of our C-V measurement technique do not support the motion of ions for explaining hysteresis. Again, the latter is more reflective of an increase in the density of deep states.

### 4. CONCLUSIONS

The transient behavior observed in C-V measurements obtained on CdTe devices deposited on either SnO2 or CTO/ZTO based substrates is likely due to the presence of deep states in the space-charge region. The increase in C-V hysteresis with stress testing of these cells is likely due to an increase in deep state density. This increases space-charge recombination. Higher levels of hysteresis are correlated with higher levels of degradation in cells. The latter is correlated with higher levels of recombination. Increased recombination does not appear to increase voltage-dependent collection. TID measurements performed as part of our C-V measurement technique do not support the motion of ions for explaining hysteresis. Again, the latter is more reflective of an increase in the density of deep states.

### 5. ACKNOWLEDGEMENTS

This work was supported by the U.S. Department of Energy under Contract No. DOE-AC36-08G028308 with the National Renewable Energy Laboratory (NREL). The authors thank Dr. Jian Li, and Clay Dehart for modifying the Labview code used for the capacitance and J-V measurements respectively.

### 6. REFERENCES

## REPORT DOCUMENTATION PAGE

<table>
<thead>
<tr>
<th>1. REPORT DATE (DD-MM-YYYY)</th>
<th>March 2011</th>
</tr>
</thead>
<tbody>
<tr>
<td>2. REPORT TYPE</td>
<td>Conference Paper</td>
</tr>
<tr>
<td>3. DATES COVERED (From - To)</td>
<td></td>
</tr>
</tbody>
</table>

### 4. TITLE AND SUBTITLE

Effect of Hysteresis on Measurements of Thin-Film Cell Performance

### 5. AUTHOR(S)

D. Albin and J. del Cueto

### 6. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES)

National Renewable Energy Laboratory  
1617 Cole Blvd.  
Golden, CO 80401-3393

### 7. DISTRIBUTION AVAILABILITY STATEMENT

National Technical Information Service  
U.S. Department of Commerce  
5285 Port Royal Road  
Springfield, VA 22161

### 14. ABSTRACT (Maximum 200 Words)

Transient or hysteresis effects in polycrystalline thin film CdS/CdTe cells are a function of pre-measurement voltage bias and whether Cu is introduced as an intentional dopant during back contact fabrication. When Cu is added, the current-density (J) vs. voltage (V) measurements performed in a reverse-to-forward voltage direction will yield higher open-circuit voltage (Voc), up to 10 mV, and smaller short-circuit current density (Jsc), by up to 2 mA/cm², relative to scanning voltage in a forward-to-reverse direction. The variation at the maximum power point, Pmax, is however small. The resulting variation in FF can be as large as 3%. When Cu is not added, hysteresis in both Voc and Jsc is negligible however Pmax hysteresis is considerably greater. This behavior corroborates observed changes in depletion width, Wd, derived from capacitance (C) vs voltage (V) scans. Measured values of Wd are always smaller in reverse-to-forward voltage scans, and conversely, larger in the forward-to-reverse voltage direction. Transient ion drift (TID) measurements performed on Cu-containing cells do not show ionic behavior suggesting that capacitance transients are more likely due to electronic capture-emission processes. J-V curve simulation using Pspice shows that increased transient capacitance during light-soak stress at 100 °C correlates with increased space-charge recombination. Voltage-dependent collection however was not observed to increase with stress in these cells.

### 15. SUBJECT TERMS

CdTe solar cell; transient capacitance; DLTS; transient ion drift; reliability; degradation; voltage-dependent collection

### 16. SECURITY CLASSIFICATION OF:

- a. REPORT Unclassified
- b. ABSTRACT Unclassified
- c. THIS PAGE Unclassified

### 19. TELEPHONE NUMBER (Include area code)

UL